Structural analysis of Ba_{0.8}Sr_{0.2}Ti_{0.6}Zr_{0.3}Mn_{0.1}O₃ ceramics

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Polycrystalline Ba_{0.8}Sr_{0.2}Ti_{0.6}Zr_{0.3}Mn_{0.1}O₃ was synthesized by solid-state reaction at 1600°C. The single phase formation of the compound without any impurities was confirmed by the X-ray diffraction technique. The prepared compound crystallized to a cubic structure with a space group of Pm-3m and the refined lattice parameters were a = b = c = 4.0253 Å, $\alpha = \beta = \gamma = 90^{\circ}$. Rietveld refinement was carried for the powder XRD data using GSAS software and the experimental data peaks were indexed by Powder X software.

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I. INTRODUCTION

Barium titanate ceramics have gained lot of attention in the scientific community due to the variation of properties due to the doping in barium or titanium sites (Levin et al., 2011). Due to its non-toxic behavior, BaTiO₃ is a better alternative to lead-based materials (Xue et al., 2011). BaTiO₃ possess superior ferroelectric behavior and has three phase transitions at different temperature regimes. At around -80° C, it transforms from rhombohedral to orthorhombic, at around 5°C, it transforms from orthorhombic to tetragonal, and at around 130°C, it transforms from tetragonal to cubic (Zhou et al., 1999). The temperature for phase transformation in BaTiO₃ shifts due to the A-site doping of Sr²⁺ or Ca²⁺ and Barlo3 sints due to the 1 site oping of Mn^{4+} , Fe^{2+} , Zr^{4+} , and Nb^{5+} making it suitable for high- or low-temperature piezoelectric device fabrication (Sindhu et al., 2013). The B-site substitution of Zirconium tunes the dielectric behavior of BaTiO₃ and increases its permittivity (Wang et al., 2014). The substitution of Mn and Zr in B-site leads to the formation of oxygen vacancy due to the charge imbalance and these oxygen vacancies induce magnetism in ceramics (Das et al., 2012).

The variation of Strontium content in BaTiO₃ shifts the transition and alters the electrical properties. Earlier reports have suggested that the doping of strontium in barium site leads to a transformation in the crystal structure from tetragonal to cubic (Sindhu et al., 2013). The smaller ionic radii of Sr^{2+} (1.44 Å) in comparison with Ba^{2+} (1.61 Å) has led to the reduction of *c/a* ratio and shifting of tetragonal structure to ideal cubic structure (Yu et al., 2015). In our sample, we also experienced the transformation of crystal structure from

tetragonal to cubic due to the increasing strontium concentration. The crystal structure for $Ba_{1-x}Sr_xTi_{0.6}Zr_{0.3}Mn_{0.1}O_3$ exhibited a tetragonal structure for x = 0 while for x = 0.2 the crystal structure transformed to an ideal cubic structure (Nandan and Kumar, 2017). Generally, $BaTiO_3$ exhibits cubic structure at around 130°C, but in our case due to the strontium doping, we were able to achieve cubic structure at room temperature. This raised our interest to analyze the powder X-ray diffraction (XRD) data and herein we are reporting the powder XRD pattern for BSTO samples.

II. EXPERIMENTAL

A. Synthesis

Polycrystalline samples of Ba_{0.8}Sr_{0.2}Ti_{0.6}Zr_{0.3}Mn_{0.1}O₃ (BSTO) were prepared by a standard solid-state reaction method. Initially, precursors barium carbonate, strontium carbonate, titanium dioxide, zirconium oxide, and manganese oxides were weighed in stoichiometric ratios and mixed by a ball milling process. The homogeneous mixture was calcined up to 1550°C in air atmosphere with intermediate grindings. The experimental procedures are detailed in our previous report (Nandan and Kumar, 2017).

B. Data collection

Powder XRD data of the samples were measured using a Bruker D8 Advance (Germany) diffractometer. The sintered powders were ground and loaded in a zero background (911) Si single-crystal wafer holder. The instrument was operated in Bragg-Brentano geometry with fixed slits. The diffraction data for the sample was recorded using Cu *K*-alpha-1 ($\lambda = 1.54060$ Å), *K*-alpha-2 ($\lambda = 1.54439$ Å), and *K*-beta ($\lambda = 1.39222$ Å) radiation as the source which is operated at a

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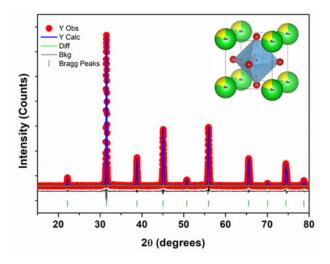


Figure 1. Rietveld refinement of powder XRD data of BSTO polycrystalline sample. The inset shows the BSTO crystal structure.

TABLE I. Fractional coordinates for BSTO samples.

Atoms	x	у	z	Occ	$U_{\rm iso}$	
Ba	0	0	0	0.8	0.0133 (20)	
Sr	0	0	0	0.2	0.0133 (20)	
Ti	0.5	0.5	0.5	0.6	0.027 (3)	
Zr	0.5	0.5	0.5	0.3	0.027 (3)	
Mn	0.5	0.5	0.5	0.1	0.027 (3)	
0	0	0.5	0.5	1	0.103 (8)	

TABLE II. Crystal data from GSAS program and Powder X software.

Crystal data from GSAS Program	Crystal data from Powder X software		
Lattice parameters: a = b = c = 4.0253 (30) Å $\alpha = \beta = \gamma = 90^{\circ}$ Volume = 65.22 (15) Å ³ Z = 3 Space group: <i>Pm</i> -3 <i>m</i> Crystal structure: Cubic $\chi^2 = 1.873$ $R_{wp} = 7.597$	Lattice parameters: a = b = c = 4.0321 (11) Å $\alpha = \beta = \gamma = 90^{\circ}$ Volume = 65.55 Å ³ Space group: <i>Pm</i> -3m Crystal structure: Cubic		

voltage of 40 kV and a current of 30 mA with a goniometer radius of 217.5 mm. The data were recorded at a 2θ range from 10° to 80° with a step size of 0.015°.

TABLE III. Powder diffraction data of BSTO polycrystalline material.

TABLE IV. Selected bond lengths for BSTO samples.

Metal-oxygen	Bond length (Å		
Ti–O	2.01265		
Zr–O	2.01265		
Mn–O	2.01265		
Ba–O	2.84632		
Sr–O	2.84632		

III. RESULTS

The refined powder XRD pattern for the BSTO samples is shown in Figure 1. Rietveld refinement for the diffraction data were carried by using GSAS-II program (Toby and Von Dreele, 2013). The initial structural parameters for refinement were taken from previous reports (Kim et al., 2004). The background was modeled using a Chebyshev polynomial. The fixed positional and isotropic displacement parameters were constrained for Ba and Sr to be the same and Ti, Zr, and Mn to be the same. Site occupancies were set and held fixed to the values determined by weights of the various starting materials mixed together and calcined given that no impurity phases evolved, even when sintered at higher temperatures. The powder data fitted well for Pm-3m space group and the fractional coordinates are shown in Table I. The results of the refinement (lattice parameter and R factors) are shown in Table II. The crystal structure of BSTO has been elucidated by Vesta software and shown as the inset in Figure 1 (Momma and Izumi, 2013). The experimental data were also indexed by Powder X software (Dong, 1999) and the results of the refined parameters are shown in Tables II and III. K-alpha-2 stripping has been done before peak indexing in Powder X software. The various metal-oxygen bond lengths from the refined data are given in Table IV.

IV. CONCLUSION

Polycrystalline samples of BSTO are prepared by a conventional solid-state reaction. Powder XRD patterns for the prepared sample confirmed the single-phase formation of the compound without any impurities. The prepared sample crystallized in a cubic structure (*Pm*₇3*m* space group) with lattice parameters a=b=c=4.0253 Å, $\alpha=\beta=\gamma=90^{\circ}$. Rietveld refinement for the XRD pattern was done by GSAS program and the peak indexing was carried out by Powder X software.

S. No	$2\theta_{\rm exp}$	$2\theta_{\mathrm{cal}}$	$2\theta_{\rm diff}$	d_{\exp}	$d_{ m calc}$	Intensity (exp)	h	k	l
1	22.146	22.161	-0.015	4.01077	4.02600	44	1	0	0
2	31.470	31.480	-0.01	2.84043	2.84681	1000	1	1	0
3	38.767	38.757	0.01	2.32093	2.32441	176	1	1	1
4	45.047	44.997	0.05	2.01088	2.013	372	2	0	0
5	50.701	50.660	0.041	1.79911	1.80048	38	2	1	0
6	55.926	55.895	0.031	1.64278	1.64361	433	2	1	1
7	65.535	65.526	0.009	1.42324	1.42341	205	2	2	0
8	70.049	70.058	-0.009	1.34215	1.342	19	2	2	1
9	74.451	74.464	-0.013	1.27332	1.27313	165	3	1	0
10	78.754	78.777	-0.023	1.21417	1.21388	41	3	1	1

V. DEPOSITED DATA

The Crystallographic Information Framework (CIF) file was deposited with the ICDD. The data can be requested at pdj@icdd.com.

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