# STUDY OF LOW-PRESSURE ARGON ADSORPTION ON SYNTHETIC NONTRONITE: IMPLICATIONS FOR SMECTITE CRYSTAL GROWTH

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Abstract—Because relatively little information about the crystal-growth process of smectite is available, the process was assessed here by studying the size and shape of nontronite particles synthesized at six different temperatures from 75 to 150ºC over a period of 4 weeks.

The morphology of nontronites was studied using low-pressure isotherms of argon adsorption at 77 K, a method which enables the measurement of the basal and edge surface areas of the nontronite particles and of their mean diameter and thickness. During the crystal growth of nontronite, the mean particle length increased whereas their thickness (and the number of stacked layers) did not vary significantly.

A specific two-dimensional crystal-growth process was observed for smectite via the lateral extension of the layers. This process also appears to occur during the growth of neoformed natural smectite.

Key Words—Argon Adsorption, Crystal Growth, Crystallinity, Infrared Spectroscopy, Isotherm, Nontronite, Particle Size, Smectite, Surface Area, Synthesis.

## INTRODUCTION

The most notable property of smectites, compared to other clay minerals, is the expandability of its layers, which is linked to the nature and number of polar molecules (water, glycol, etc.) that surround the interlayer cations (Brindley and Brown, 1980). The c crystallographic parameter is variable for layers with a constant chemistry. In smectite, the interlayer cations (e.g.  $Na<sup>+</sup>$ ,  $K^+$ ,  $Li^+$ ,  $Ca^{2+}$ , and  $Mg^{2+}$ ) are exchangeable and depend on the composition of the last fluid that was in contact with the smectite. Thus, only the 2:1 layer is relatively stable with time, in contrast to the labile interlayer.

Smectites are generally described as thin crumpled sheets (with a few tens of stacked layers), a few hundred nanometers wide, usually anhedral, which form aggregates (Güven, 1988). The study of the morphological evolution of smectites during crystal growth in natural or experimental systems is not straightforward, therefore. Numerous studies of smectite synthesis exist in the literature (for examples see reviews by Kloprogge et al., 1999, and Zhang et al., 2010). In such studies, however, data on the shape and size of the smectite particles are often discussed with respect to synthesis conditions (temperature, pressure, pH, ...) and not with respect to the crystal-growth issue.

Transmission electron microscopy (TEM) and atomic force microscopy (AFM) are useful tools for studying the morphology of clays and many studies have been

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devoted to imaging in three dimensions clay minerals with euhedral or sub-euhedral character, such as kaolinite, micas, and mixed-layer clays (Meunier, 2010 and references therein). For these clay minerals, crystal growth occurs via three-dimensional processes and often exhibits polygonal and spiral-growth patterns (Baronnet, 1972; Meunier, 2010). Researchers have also used TEM and AFM to study smectite dissolution (e.g. Bickmore et al., 2001; Bauer et al., 2006; Kuwahara, 2006).

The Crystal Size Distributions (CSDs) of illite and illite-smectite samples were studied by Eberl et al. (1998) and Srodon<sup> $et$  al. (2000) using TEM imaging. The</sup> CSDs may have distinctive shapes related to the mineral's crystal-growth history: nucleation, growth in an open or closed system, Ostwald ripening, etc. The CSDs method was used by Christidis (2001) to study a bentonite from Greece. Only the length and width of particles were measured, suggesting supply-controlled crystal growth in an open system or random ripening in a closed system.

An alternate approach to studying clay morphology is high-resolution, low-pressure argon adsorption, a method based on surface-energy heterogeneity analysis, using the summation of the local derivative isotherms method (Villiéras et al., 1992, 1997a, 1997b, 2002; Michot and Villiéras, 2006). The validity of this method was first tested on kaolinite and sepiolite via comparison of results obtained with those from low-temperature adsorption microcalorimetry (Villiéras et al., 1992, 1997a). More recently, the study of the geometric properties of smectite, kaolinite, and illite using both AFM and argon adsorption (Tournassat et al., 2003, Sayed Hassan et al., 2006) demonstrated the consistency

$T$ (°C)	(001)	(06,33)		
	$H(\AA)$	N	D(A)	
150	76	6.3	189	
125	63	5.2	190	
110	50	4.1	138	
100	45	3.8	142	
90	48	4.0	114	
75	nm	nm	nm	

Table 1. Coherent scattering domain size (CSDS) of nontronite particles from powder XRD data.

 $H =$  crystal coherency along the  $c^*$  direction (air dry

conditions, Na as interlayer cation);  $N =$  number of stacked layers (assuming  $d_{001} = 12$  Å);  $D =$  crystal coherency in the  $(001)$  layer plane; nm = not measurable (crystallinity too low).

between these two methods. Argon adsorption has also been used to study the shape and size of smectite particles, including saponite (Michot and Villiéras, 2002), montmorillonite, and nontronite (Tournassat et al., 2003; Peronnet et al., 2007; Le Forestier et al., 2010). High-resolution, low-pressure argon adsorption enables the simultaneous investigation of all of the clay particles in a sample, which results in the accurate determination of mean basal and edge surface areas (Sayed Hassan et al., 2006).

The objective of this study was to determine changes in particle shape and size in a series of nontronite samples as a function of synthesis temperature, by applying high-resolution, low-pressure argon adsorption.

# MATERIALS AND METHODS

#### Sample description and characterization

A series of nontronite samples was synthesized by Decarreau et al. (2008) from a coprecipitated gel with the composition  $Si<sub>2</sub>FeNa<sub>2</sub>O<sub>6</sub>$ .nH<sub>2</sub>O over a 4-week period at temperatures that ranged from 75 to 150ºC. Nontronite was demonstrated to be the only mineral that could precipitate under these synthesis conditions (Decarreau et al., 2004, 2008). These synthetic nontronites were studied in detail using X-ray diffraction (XRD), TEM, chemical analyses, differential thermal and gravimetric analysis (DTA-DTG), and Fourier-transform infrared (FTIR), Mössbauer, and X-ray absorption near edge (XANES) spectroscopies (Decarreau et al., 2008). Irrespective of the synthesis temperature, the nontronite samples exhibited the same structural formula of  $(Si_{3.25}Fe_{0.75}^{3+})Fe_2^{3+}O_{10}(OH)_2$ , Na<sub>0.75</sub>. An increase in the synthesis temperature increased the crystallinity of samples. The synthesis products consisted of nontronite and some amorphous residual phase. As determined from the DTG data (Decarreau et al., 2008), the proportion of starting gel which crystallized into nontronite increased from 70% to >90% with the increase in synthesis temperature. Taking into account the uncertainties in the DTG data, the samples obtained at 125 and 150ºC were assumed to be of pure nontronite.

From XRD powder data (Decarreau et al., 2008), the coherent scattering domain size (CSDS) (Moore and Reynolds, 1989, p. 87) was measured using the Scherrer equation ( $L = K \cdot \lambda / B \cdot \cos \theta$ ) in the (001) plane and along the  $[001]$ <sup>\*</sup> direction (also known as the  $c$ <sup>\*</sup> direction) from the widths of the (06,33) and (001) reflections, respectively (Table 1). The K values in the Scherrer equation were 1 for the (001) reflection and 1.91 for the (06,33) composite reflection (Brindley and Brown, 1980, pp. 131, 139). When the synthesis temperature was increased, the CSDS in the plane of the layers was increased at temperatures from 75 to 125ºC, and then remained constant (Figure 1). Along the c\* direction, the CSDS remained constant between 75 and 100ºC and then increased slightly (Figure 1). Using a  $d_{001}$  value of 12 Å, the number of stacked layers as determined from the XRD data was between 4 and 6.



Figure 1. Coherent scattering domain size (CSDS) (from XRD, in the layer plane (2) and in the c\* direction (3)) and dimensions (from argon adsorption: particle length (1) and thickness (4)) of particles vs. synthesis temperature of nontronites.

In the TEM images (Decarreau *et al.*, 2008), irrespective of the synthesis temperature, isolated nontronite particles appeared as crumpled flakes with rolled edges associated in micron-sized aggregates. The TEM images showed that the aggregates contained numerous particles <10 nm in size. These small particles were less evident in the samples prepared at 125 and 150ºC. The largest observed particles were 400 nm for the sample prepared at 150ºC. Even after strong ultrasonic treatment, the nontronite particles still formed thick aggregates irrespective of the sample: a measurement of the particle-size distribution was not possible.

## Low-pressure argon adsorption at 77 K

The fundamentals of the low-pressure argon adsorption approach and the derivative isotherm summation (DIS) method were given by Villiéras et al. (1992, 1997a, 1997b, 2002). The low-pressure isotherms of argon adsorption at 77 K were recorded using a lab-built, automatic, quasi-equilibrium volumetric apparatus (Michot et al., 1990; Villiéras et al., 1992, 1997b). Approximately 1 g of the sample was outgassed overnight at  $110^{\circ}$ C under a residual pressure of  $10^{-4}$  Pa. After the samples were outgassed, a slow and constant flow of argon was introduced into the adsorption cell through a micro-leak up to the Brunauer-Emmet-Teller (BET) domain ( $P/P_0 \ge 0.15$ ). If the flow rate is sufficiently low, the measured pressures are considered to be quasi-equilibrium pressures (in the range of  $10^{-3}$  to  $3 \times 10^4$  Pa). From the recording of the quasi-equilibrium pressure as a function of time, high-resolution adsorption isotherms were obtained for the apparent filling of the first monolayer ( $\pm$  BET Monolayer capacity, 0.08  $\geq$  $P/P_0 \ge 0.15$ ) with >2000 experimental data points. The derivative of the adsorbed quantity is calculated as a function of the logarithm of the relative pressure,  $ln(P/P_0)$ , which corresponds to the free energy of adsorption in kT or RT units (Villiéras et al., 1992, 1997a). The total derivative adsorption isotherm on a heterogeneous surface is then simulated by the sum of the local theoretical derivative adsorption isotherms using concepts that describe the adsorption phenomena on heterogeneous surfaces and the DIS procedure (Villiéras et al., 1992, 1997a). For each local isotherm, the BET multilayer adsorption model was used and the following parameters obtained: o, the lateral interaction between two neighboring adsorbed molecules;  $\ln(P/P_0)$ , the position of the peak; and Vm, the monolayer capacity. The DIS fitting procedure has been described extensively by Villiéras et al. (1992, 1997a, 1997b, 2002) and, for swelling clays, by Perronnet et al. (2007).

For clay minerals with large cations (e.g.  $K^+$ ) on their external surfaces, the values of the basal surface and edge surface areas measured by low-pressure argon adsorption can be significantly smaller as shown by Sayed-Hassan et al. (2005) for kaolinite edge surfaces and by Bardot et al. (1998) for illite basal and edge

surfaces. Therefore, prior to the adsorption experiments, the clay samples were  $Na^+$ -saturated using a 1 M NaCl solution and washed until all excess ions were removed, as indicated by the  $AgNO<sub>3</sub>$  test.

## RESULTS

Irrespective of the synthesis temperature, the derivative isotherms of the low-pressure argon adsorption of the synthesized nontronites exhibited the same features (Figure 2). As typical for phyllosilicates (Sayed Hassan et al., 2006; Villiéras et al., 1992, 1997a, 1997b), the derivative isotherms exhibited a peak at  $ln(P/P_0)$  near  $-4$  to  $-5$ , and a shoulder and a tail at greater energy (*i.e.* at lower values of  $ln(P/P_0)$ . For kaolinite, the main peak at low energy could be assigned to the adsorption on basal faces while the shoulder at higher energy and highenergy adsorption peaks could be assigned to the adsorption on edge faces (Villiéras et al., 1992). Further studies demonstrated that the DIS peak attribution for kaolinite could be applied to argon adsorption on other clay minerals including smectites  $(e.g.$  Villieras  $et$ al., 1992; Perronnet et al., 2007; Sayed Hassan et al., 2006).

Each derivative adsorption isotherm could be fitted using the DIS procedure (Villiéras et al., 1997a; Bardot et al., 1998; Sayed Hassan et al., 2005, 2006; Perronnet et al., 2007) with five characteristic adsorption domains (Figure 3, Table 2). The parameters of the DIS procedure were similar for the samples that were synthesized at 125 and 150ºC (Table 2). In the present case, the total derivative isotherms for clay minerals could be modeled using three BET local-derivative isotherms at greater energies (domains 3-5 in Table 2) which were attributed to the adsorption on the edge surfaces, and two local derivative isotherms at lower energies (domains 1 and 2 in Table 2) which were attributed to adsorption on the basal surfaces.

The derivative curve of the starting gel (Figure 2) differed substantially from those of the nontronites synthesized. Most notably, the derivative curve of the starting gel did not exhibit a clear peak at  $ln(P/P_0)$  at approximately  $-4$  to  $-5$ , *i.e.* characteristic of poorly crystallized solids, such as amorphous silica (Villiéras et al., 1998; Rudzinski et al., 1999). Observations by TEM of similar gels revealed that they contain rounded particles, with diameters of  $\sim$ 200 A, organized as diffuse aggregates (Decarreau et al., 1987). The measured total surface area (TSA) of the gel was  $127 \text{ m}^2/\text{g}$  (Table 3), a value corresponding to spherical particles with a diameter of  $160$  Å, in reasonable agreement with the TEM results. Thus, the edge surface area (ESA) and basal surface area (BSA) estimates from argon adsorption were irrelevant for the starting gel (Table 3).

For all of the other samples, the peak positions at  $\ln(P/P_0) \approx -5$  and at  $\ln(P/P_0) \approx -7$  were typical of basal and edge surfaces with  $Na<sup>+</sup>$  as surface-exchange-



Figure 2. Derivative isotherms of low-pressure argon adsorption at 77 K of synthetic nontronites. The numbers on the key refer to the synthesis temperature (ºC). Gel: starting gel.

able cations (Bardot et al., 1998), as expected given the sample-preparation procedure. The evolution of the lateral interaction parameter o/kt for domain 2, which ranged from 0.5 for the starting gel to 1.4 for the 150ºC sample, indicated clearly that the surface homogeneity for argon increased with increasing synthesis temperature. Following Villiéras et al. (1997b) and Perronnet et al. (2007), the number of surface defects or the surface roughness on the basal surfaces of the nontronite samples were assumed to decrease, therefore.

The BSA, ESA, and TSA were calculated on the basis of a 13.8  $\AA^2$  cross-sectional area for adsorbed argon (Table 3).

Nontronite particles were treated as circular disks to estimate the mean dimensions. The thickness  $(H)$  and particle length  $(D)$  of the particles (Table 3) were calculated from the BSA and ESA using the equations:

$$
H = 2/BSA \cdot \rho \text{ and } D = 4/ESA \cdot \rho \tag{1}
$$

where  $\rho$  is the particle density (3.0 g/cm<sup>3</sup> according to the structural formula). Several clear trends appeared



Figure 3. Experimental derivative argon adsorption obtained at 77 K for the nontronite synthesized at 150ºC and its decomposition using the derivative isotherm summation (DIS) method. Solid line: experimental; dotted line: fit; dashed lines: adsorption domains obtained using the DIS procedure.

	DIS domain				Synthesis temperature $(^{\circ}C)$			
	parameters	Starting gel	75	90	100	110	125	150
Domain 1	$ln(P/P_0)$	$-2.61$	$-2.70$	$-2.61$	$-2.58$	$-2.55$	$-2.40$	$-2.46$
	$Vm$ (cm <sup>3</sup> /g)	13.6	21.8	25.1	23.4	19.9	15.0	16.7
	$\omega/KT$	$\theta$	$\theta$	$\theta$	$\theta$	$\theta$	$\theta$	$\theta$
Domain 2	$ln(P/P_0)$	$-5.04$	$-4.95$	$-5.01$	$-4.98$	$-4.98$	$-4.89$	$-4.92$
	$Vm$ (cm <sup>3</sup> /g)	7.9	19.6	23.8	24.5	24.4	23.1	22.1
	$\omega/KT$	0.5	1.1	1.1	1.2	1.1	1.2	1.4
Domain 3	$ln(P/P_0)$	$-6.93$	$-6.84$	$-6.93$	$-6.93$	$-7.05$	$-7.44$	$-7.29$
	$Vm$ (cm <sup>3</sup> /g)	8.2	23.9	24.7	22.6	20.1	14.8	14.8
	$\omega/KT$	$\theta$	$\theta$	$\theta$	$\Omega$	$\theta$	$-0.2$	$-0.3$
Domain 4	$ln(P/P_0)$	$-9.21$	$-9.48$	$-9.84$	$-9.78$	$-9.87$	$-10.77$	$-10.77$
	$Vm$ (cm <sup>3</sup> /g)	3.4	9.9	9.7	7.8	7.1	4.3	3.7
	$\omega K T$	$\Omega$	$\theta$	$\Omega$	$\Omega$	$\theta$	$\Omega$	0.2
Domain 5	$ln(P/P_0)$	$-11.91$	$-12.21$	$-12.27$	$-12.24$	$-12.42$	$-13.02$	$-13.14$
	$V\text{m}$ (cm <sup>3</sup> /g)	1.1	8.1	6.6	6.2	5.3	2.3	2.2
	$\omega/KT$	$\Omega$	$\theta$	$\Omega$	$\Omega$	$\theta$	$\Omega$	$\theta$

Table 2. Main parameters obtained from the application of the DIS method to argon adsorption at 77 K on synthesized nontronites.

 $ln(P/P_0)$ : peak position in logarithm of the relative argon pressure unit;  $Vm$  (cm<sup>3</sup>/g): monolayer capacity; o/KT: lateral interaction between two neighboring adsorbed molecules.

from these data. The synthesis temperature affected the size and shape of the nontronite particles (Table 3). The TSA decreased from 75 to 125ºC and remained fixed between 125 and 150ºC (Figure 4). The decrease in the TSA was not related to the last appearance of the gel when the synthesis temperature was increased. In fact, the last appearance of the gel would be expected to promote an increase in the TSA. The observed decrease in the TSA was related to an increase in the particle size of the smectites synthesized (Decarreau, 1980; Decarreau and Bonnin, 1986; Decarreau et al., 1987). In contrast, the ESA and BSA (expressed as % of the TSA) exhibited opposite trends (Figure 5): between 75 and 125ºC the BSA increased whereas the ESA decreased, and the ESA and BSA did not change between 125 and 150ºC. These trends are related to an increase in the particle length, although the thickness did not change significantly (Figure 1). The

mean diameter of the synthetic nontronites increased from 86 to 173 Å between 75 and 125 °C and did not increase between 125 and 150ºC. For all of the synthesis temperatures, the mean thickness of the particles remained nearly constant at 40 Å. The  $d_{001}$  value was 10 Å under the argon-adsorption experimental conditions. Under these conditions, the mean number of stacked layers for the synthesized nontronites was four, irrespective of the synthesis temperature.

For the nontronite samples that were synthesized at 75 and 100ºC, the maximum amount of remaining gel was 30% (Decarreau et al., 2008). Although the TSA of the gel was low, the TSA of the nontronite particles was slightly larger because the TSA was measured for the entire sample. The derivative adsorption isotherm of the starting gel was fitted by the DIS method using the five absorption domains similar to those of nontronite, in

Table 3. Basal surface area (BSA), edge surface area (ESA), total surface area (TSA)  $(m^2/g)$ , thickness (H), and particle length (D) of synthesized nontronite particles obtained from low-pressure argon adsorption data. Values given in italics are BSA and ESA expressed as a percentage of TSA.

Synthesis temperature $(^{\circ}C)$	Starting gel	75	90	100	110	125	150
<b>BSA</b>		154 50	182 53	178 56	164 57	142 64	143 65
<b>ESA</b>		155 50	152 47	136 44	121 43	79 36	77 35
TSA	127	309	333	313	285	221	220
H		43	37	37	41	47	47
D		86	88	98	110	169	173



Figure 4. Total surface area  $(TSA)$   $(m^2/g)$  of nontronite particles (from argon adsorption) vs. synthesis temperature.

order to evaluate the gel contribution in argon adsorption on the whole sample (Table 2). The measured TSA for each sample was then equal to the sum of the nontronite TSA and starting-gel TSA. The actual values of the TSA, ESA, and BSA of nontronite could be obtained, using the gel:nontronite weight ratio determined from DTG data and using the data in Table 2, for the five adsorption domains. Using a mixing model, the TSA of nontronite would increase but the ESA/BSA ratio would not change (not shown). This mixing model could not be applied quantitatively because the gel:nontronite weight ratio, as measured by DTG, was not sufficiently precise, and the TSA of the remaining untransformed product might have been different from that of the starting gel and might differ from one sample to another.

#### DISCUSSION

#### Consistency of data

The XRD data and the argon adsorption data followed the same trend. The parameters measured were in the same range but were not equal (Tables 1, 3; Figure 1).

Both data sets showed an increase in the particle length and CSDS in the (001) plane of nontronite



Figure 5. Relative (% of total surface area, TSA) ratio of edge surface area  $(ESA)$  and basal surface area  $(BSA)$  of nontronite particles (from argon adsorption) vs. synthesis temperature.

particles synthesized at temperatures from 75 to 125ºC, and these values remained constant for samples prepared at 150ºC. The CSDS that were measured by XRD are slightly greater than the mean particle lengths that were measured by argon adsorption. The XRD data gave the mean length of the crystallographic lattice without any defects (single crystal). The CSDS, in the (001) plane, as measured by XRD, cannot be greater than the particle length of nontronite. The differences between the XRD and argon adsorption data are probably explained by the uncertainties in the measurements.

According to both data sets, the mean number of stacked layers (N) in nontronite was four for samples synthesized at temperatures from 75 to 110ºC. For samples prepared at 125 to 150°C, N increased slightly (by <1 layer) according to the argon data and was six according to the powder XRD data. The number of stacked layers, as measured by argon adsorption, was apparently four for all samples. The discrepancy between the XRD and argon adsorption data may be related to inter-particle diffraction (Eberl et al., 1998) for nontronites prepared at higher temperatures. Argon adsorption appears to reflect the actual size and shape of the nontronite particles better than the values obtained from the XRD data.

## Crystal growth of smectites

Synthesized nontronite crystal growth. Because nontronite is the only mineral phase that can precipitate under the experimental conditions used (Decarreau et al., 2008; Andrieux and Petit, 2010), the chemical evolution of the system involves only the nucleation and growth of nontronite. The influences of the synthesis duration and temperature on similar nontronite crystallization were studied by Decarreau (1980) and Decarreau *et al.* (1987). The smectite crystal size within the (001) plane was time dependent, and the kinetics of crystal growth followed the Arrhenius equation (Carrado et al., 2006). The mean particle length  $(D)$ , within the  $(001)$  plane was:

$$
D = A \cdot \exp(-E/RT) \cdot t
$$

where  $t$  is time and  $T$  is absolute temperature. For the present experiments, time and temperature played analogous roles. Syntheses performed at increased temperatures with a fixed aging time were analogous to syntheses performed at a fixed temperature with increased aging times. In the present work, as the synthesis temperature increased, the number of stacked layers did not increase significantly, although the mean length of the particles doubled. This observation suggests that the crystal growth was dominated by the lateral extension of the synthesized nontronite layers. This lateral extension measured for the synthetic nontronites is a crystal-growth process that differs from those usually described for clay minerals that are more three-dimensional, such as kaolinite, chlorite, and mica. Polygonal and/or broadly circular spiral-growth

patterns have often been observed on the growth surfaces of biotite (Amelinks, 1952), phlogopite (Baronnet, 1972), illite (Inoue and Kitagawa, 1994; Kitagawa, 1998), rectorite (Kitagawa, 1997), kaolinite (Sunagawa and Koshino, 1975), and mixed-layer illite-smectite (Kitagawa and Matsuda, 1992). Two-dimensional nucleation was observed on the (001) growth surfaces of sudoite (Jige et al., 2003), which produced an increase in the number of stacked layers. The latter crystal-growth process is less commonly observed because a greater degree of fluid oversaturation is required.

Using the periodic bond chain (PBC) theory of Hartman (1973), a large number of defects in clay minerals was suggested by Meunier (2010) to disrupt the PBCs, thus precluding crystal growth. The PBCs present in this study were those within the (001) plane, i.e. the PBCs of  $[100]$ ,  $[110]$ , and  $[110]$ . Following Meunier (2010), the PBC theory of Hartman (1973) suggests implicitly a crystal-growth process of clay minerals essentially bi-dimensional within the (001) plane, which was observed for the synthetic nontronites here.

During crystal growth, an increase in the particle size entailed a decrease in the TSA per unit mass. For a constant number of stacked layers in the clay particles, the BSA per unit mass is constant during crystal growth  $(H = constant in relation 1)$ , and a decrease in the TSA is only due to a decrease in the ESA. This relation was observed for the synthesized nontronites (Table 3): the BSA fluctuated around a mean value of 160  $\text{m}^2/\text{g}$ , whereas the ESA decreased continuously from 115 to 77 m<sup>2</sup>/g. The theoretical values of the  $TSA$  for circular particles, having a constant 40 Å thickness, were calculated as a function of the particle diameter (Figure 6). The measured TSA of the synthetic nontronites, obtained by the DIS procedure from the argon adsorption curves, fitted the theoretical curve well (Figure 6).



Figure 6. Total surface area  $(TSA)$   $(m^2/g)$  of nontronite particles vs. their mean particle length. Solid line: theoretical TSA calculated for circular particles with a constant thickness of 40 A. Open diamonds: TSA values measured from argon adsorption.

The work above demonstrated that the crystal growth of the synthetic nontronites occurred via the addition of material ('building blocks') to the edges of particles, mainly. These 'building blocks' were probably less than four layers thick but their structure could not be determined in detail. Consequently, the stacked layers might not have the same planar extension, inducing local strength. This crystal growth may explain the rolled edges of the particles as observed by TEM (Decarreau et al., 2008).

The crystal growth of the nontronites was similar for samples prepared at 125 and 150ºC. Because nontronite becomes unstable and aegirine occurs at higher temperatures (Decarreau et al. 2004), long synthesis times (i.e. several years, Decarreau et al., 1987) would be required to obtain larger crystals experimentally.

Effect of particle size on OH-stretching vibrations. The v OH bandwidth in infrared spectra is partially linked to the size of particles due to the ''crystallinity'' effect (Petit et al., 2004, 2008). A decrease in the v  $Fe<sub>2</sub><sup>3+</sup>-OH$ band width in IR spectra of synthetic nontronites was noted by Decarreau et al. (2008) when the synthesis temperature increased. The plot of the  $v Fe<sub>2</sub><sup>3+</sup>-OH$  band width vs. the mean particle length, obtained from argon adsorption, revealed that the band width reached a plateau at 45 cm<sup>-1</sup> from a particle length of  $\sim$ 180 Å (Figure 7).

Comparison with other synthetic smectites. The crystal growth of plate-shaped smectite via the lateral extension of layers is in agreement with the data from Decarreau (1980) and Decarreau et al. (1987). In these previous studies, the CSDS of synthetic Mg-smectite (stevensite) and of various synthetic nontronites were measured. The authors observed a small increase in the number of stacked layers (from two to six) and an increase along the (001) plane with an increase in the synthesis time or temperature.



Figure 7. Full width at half maximum (FWHM) of the  $v Fe<sub>2</sub><sup>3+</sup>$ -OH band observed on the MIR spectra of synthetic nontronites vs. the mean length of the particles.

Synthetic saponite (Michot and Villiéras, 2002) and montmorillonite (Le Forestier et al., 2010) were also studied using low-pressure argon adsorption. Syntheses were performed for four weeks at 400ºC at 1 kbar (various charges) for the saponite series and at 350ºC and 1.2 kbar for montmorillonite. The derivative adsorption curves were similar to those of the synthetic nontronite presented here. The saponite particles were 200-600 nm wide (mean of 315 nm) and 17-39 nm thick (mean of 25 nm). These synthetic saponite crystals were ~20 times greater in the lateral dimension and only six times thicker than the largest nontronites synthesized here. The montmorillonite synthesized was 65 nm inplane and 12 nm thick. Unfortunately, these synthetic saponite samples and montmorillonite were obtained for only one aging time and only one temperature; consequently, the crystal-growth process could not be deduced. Nevertheless, even for synthesis temperatures >300ºC, synthetic smectites contained 12-39 stacked layers which resulted in a thickness much smaller than the plane extension.

Despite the high synthesis temperature, the montmorillonite particles were only three to four times larger than the nontronite particles studied, and had a similar H/D ratio. For dioctahedral smectites (e.g. montmorillonite and nontronite) a moderate increase in synthesis temperature (or time) did not result in a significant increase in the particles size. In contrast, large smectite crystals were synthesized under extreme conditions by Nakasawa et al. (1992).

Trioctahedral smectite particles were shown by Decarreau (1980, 1983) and Decarreau et al. (1987) to be generally larger than dioctahedral smectite particles in samples prepared under similar synthesis conditions. Accordingly, argon adsorption measurements revealed that saponite particles were significantly larger (ten times) than the montmorillonite particles that were synthesized under similar conditions (Michot and Villiéras, 2002; Le Forestier et al., 2010).

Syntheses performed in the temperature range of weathering or hydrothermal systems indicated that the small size of dioctahedral smectites was constrained by the crystal-growth processes.

Comparison with natural smectites. The particle morphologies of various natural smectites have been studied using low-pressure argon adsorption, including those of MX80 montmorillonite (Tournassat et al., 2003), two montmorillonites and one nontronite  $(Fe<sup>3+</sup>$ rich beidellite) (Perronnet et al., 2007), and SWy2 montmorillonite samples (Le Forestier et al., 2010). These smectites were derived from bentonites, which resulted from the interaction of volcanic ash falls and ocean water (Moll, 2001; Christidis and Huff, 2009). From the oxygen and hydrogen isotopic data (Arslan et al., 2010 and references therein), bentonites were determined to form at relatively low temperatures

 $(\leq 100^{\circ}C)$ . The particles of these natural smectites have a size range similar to those of synthetic dioctahedral smectites, though the MX80 and SWy2 smectites contained additional stacked layers (28 nm and 34 nm thick, respectively) compared to synthetic smectites  $(11-15 \text{ nm thick})$  according to Perronnet *et al.* (2007).

The data that were obtained for natural bentonites are in agreement with the data from experimental systems (Meunier, 2006), suggesting similar crystal-growth processes.

Clay minerals that form at the Earth's surface and under sub-surface conditions exhibit a small particle size, whereas the associated minerals are generally not as small. The PBC theory (Hartman, 1973) was suggested by Meunier (2006) to be applicable to clay minerals, thus precluding their crystal growth. Implicitly, the crystal growth of clay minerals is essentially two-dimensional (Meunier, 2006) (see above) as demonstrated here both for natural and synthetic smectites.

Smectite crystal-growth mechanisms. As far as the authors are aware, no data on the crystal-growth mechanisms of smectites are available in the literature. The only permanent portion of the smectite structure is the layer because the interlayer can vary in both size and chemistry, depending on the chemistry of the surrounding fluid. Consequently, crystal-growth mechanisms in three dimensions (e.g. polygonal and spiral-growth patterns) are unlikely to occur for smectite. The crystal-growth pattern that was determined for synthetic nontronite, i.e. the lateral growth of the layer only, probably occurs for synthetic and natural dioctahedral smectites. This conclusion is suggested by the similarity of the morphologies of natural and synthetic dioctahedral smectite. For synthetic nontronite, the number of stacked layers (i.e. four) can crystallize during the nucleation stage of crystal growth or slightly afterward; smectites that were synthesized over a few days at 25ºC exhibited a broad 001 XRD reflection due to three or four stacked layers (Decarreau, 1980, 1983; Decarreau and Bonnin, 1986; Decarreau et al., 1987). For natural smectite and smectites that were synthesized at higher temperatures, particles with 10 to 40 stacked layers may represent another crystal-growth process. The AFM images of the MX 80 sample (Tournassat et al., 2003) showed broadly circular "islands" on the  $(001)$  surfaces of isolated particles 10-200 nm wide. These islands were one layer thick. This feature was consistent with the close-step pattern of two-dimensional growth, which led to an increase in the number of stacked layers. Such a growth pattern was observed for metamorphic mica (Sunagawa and Koshino, 1975) and hydrothermal sudoite (Jige et al., 2003). However, only two-dimensional crystal-growth processes occur during smectite formation. The remaining questions are as follows: what is the crystal-growth unit (building block) and does this

unit include an interlayer cation? For epitaxial growth of trioctahedral clay, nucleation began via the formation of a hydroxide sheet on which silica was adsorbed (Manceau et al., 1999; Rainer et al., 2002; Schegel et  $al., 2001$ ). This model, proposed by Caillère et al. (1956, 1957) and often used since, was based on the similarity of structures of trioctahedral Mg, Ni, and Co hydroxides and the analogous octahedral sheet of trioctahedral clay minerals. The process was different for dioctahedral structures (Decarreau, 1983). The building blocks were demonstrated to be [Si tetrahedra-Al (or Fe) octahedra-Si tetrahedra] 'monomers' for TOT dioctahedral clays (Siffert, 1962). These monomers might be the smaller building block for the crystal growth of dioctahedral TOT clays (White and Zelazny, 1988).

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