

Analytical microscopy studies of nitrogen solubility in austenite and ferrite upon welding of hyper duplex steel

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Hyper-duplex stainless steel and gas-shielded tungsten arc welding (GTAW) have been important research fields in technology and steel industry in recent times [1-4]. The effects of the addition of nitrogen during welding on the balance of the austenite/ferrite volume fraction ratio were studied in this work. The welding technique used was arc welding with tungsten gas (GTAW). The welded joints in tubes UNS S32707 of hyper-duplex steel were produced using a mixing of argon and nitrogen gas, with nitrogen added in ratios of 1.5 to 5.5%. Microstructural characterization of this material under the different processing condition was conducted by means of optical microscopy, scanning electron microscopy (SEM), local chemical analysis by X-ray energy dispersive spectroscopy (EDS), electron backscatter diffraction (EBSD) and Vickers micro hardness test. All welding conditions resulted in welded joints with equiaxial grains microstructure containing austenite both at the boundaries and in the ferrite matrix. The use of welding gases with higher percentages of nitrogen resulted in increases in the austenite volume fraction in the order of 39 to 57% when considering the nitrogen content ranging from 1.5 to 5.5%. The material used for this work was hyper-duplex tubes with 19.05 mm in diameter and 2.24 mm in thickness. Those tubes were welded in our laboratory in an automated manner using the gas tungsten arc welding (GTAW) process. The increase of the nitrogen content used in the shielding gas had no significant effect in the microhardness of the welded joints [1]. FIG.1 shows a focused ion beam (FIB) micrograph of the sample welded with 1.5% of N content in the gas mixing. The channeling contrast induced by FIB allows observing the difference of orientation between the grains of the sample, different contrast for different orientations. The EBSD results showed that the fraction of austenite increases slightly with the increase in nitrogen. The results of the phase maps obtained by EBSD and the chemical maps obtained by EDS, allows to observe that nitrogen is found in higher concentration in the austenitic phase as well as nickel, which is a well-known as austenite stabilizer (FIG.2). The phase map of FIG.2 shows the austenite (Iron Gamma) in green color. TSL software was used to collect and analyze the EBSD results.

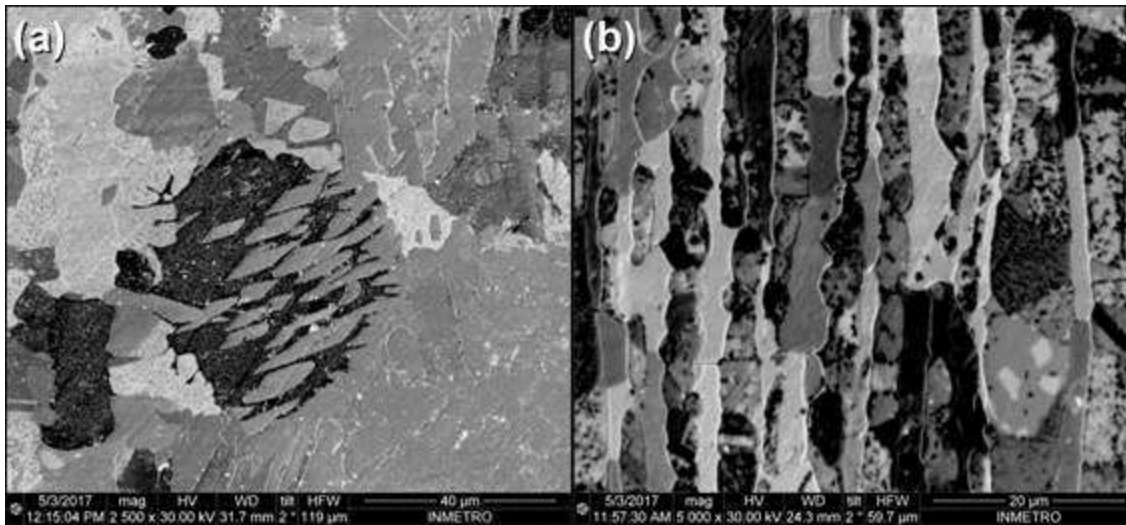


Figure 1. FIB image showing channeling contrast by ion beam of the sample welded with 1.5% of N content: (a) general region, (b) melted region. Different contrasts represent different orientations.

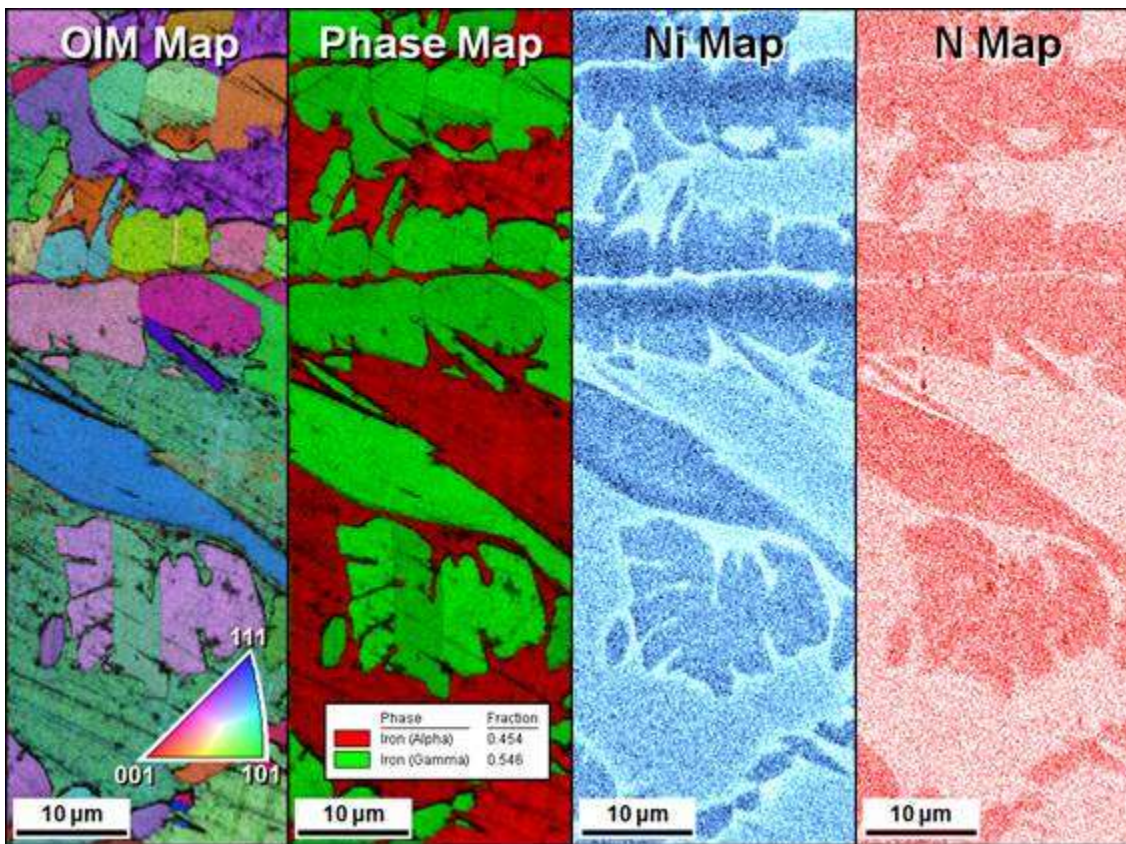


Figure 2. EBSD and chemical EDS maps results of the sample welded with 1.5% of N content: Orientation map (OIM), phase map, nickel map (Ni) and nitrogen map (N). Note the concentration of nickel and nitrogen in the austenite (green phase on the phase map).

References

- 1 - Fonseca GS, Oliveira PM, Diniz MG, Bubnoff DV, Castro JA. Sigma phase in superduplex stainless steel: formation kinetics and microstructural path. *Materials Research*. 2017;20(1):249-255. <http://dx.doi.org/10.1590/1980-5373-mr-2016-0436>.
- 2 - Kobayashi DY, Wolyneć S. Evaluation of the low corrosion resistant phase formed during the sigma phase precipitation in duplex stainless steels. *Materials Research*. 1999;2(4):239-247. <http://dx.doi.org/10.1590/S1516-14391999000400002>.
- 3 - Tavares SSM, Pardal JM, Almeida BB, Mendes MT, Freire JLF, Vidal AC. Failure of superduplex stainless steel flange due to inadequate microstructure and fabrication process. *Engineering Failure Analysis*. 2018;84:1-10. <http://dx.doi.org/10.1016/j.engfailanal.2017.10.007>
- 4 - Pimenta AR, Diniz MG, Perez G, Solorzano-Naranjo IG. Nitrogen addition to the shielding gas for welding hyper-duplex stainless steel. *Soldagem&Inspeção*. 2019;25:e2512. <https://doi.org/10.1590/0104-9224/SI25.12,A>