Aberration Corrected Lorentz Microscopy to Investigate Magnetic Domain Walls in Co-Pt Nano-Chessboards

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Lorentz Transmission Electron Microscopy (LTEM) is a technique used to image and characterize magnetic domains and magnetic domain walls. In LTEM mode, since the sample sits inside the objective lens, the lens is turned off in order to preserve the magnetic structure of the sample. A separate Lorentz lens is used to image the sample, albeit with in a great reduction in attainable magnification. The conventional LTEM offers a spatial resolution of ~10-15 nm and requires a very high defocus value (of the order of several microns) to make magnetic domain walls visible, which further degrades the resolution of the final images; this makes it challenging to study nano-scale magnetic structures. However, recent breakthroughs in aberration corrected microscopy have enabled the study of complex and fine scale magnetic domains in the Lorentz TEM mode. In this study, we use an FEI Titan 80-300 microscope with a dedicated Lorentz pole piece that offers C_S-corrected field free imaging (<2 Oe) with <1 nm resolution. The Fresnel or defocus mode of Lorentz imaging is employed to investigate domain walls. The Fresnel mode LTEM only provides qualitative information about the position of domain walls. Quantitative information about the magnetic induction can be extracted from a Fresnel throughfocus series of images (in-focus, over-focus, and under-focus) by reconstructing the phase of the electron waves using the transport-of-intensity equation. Aberration correction also reduces the defocus value required to image the domain walls and makes Fresnel images useful for quantitative analysis. A detailed description of Lorentz Fresnel mode and phase reconstruction can be found in [1].

Co-Pt ordered alloys of near-eutectoid compositions (~Co₄₀Pt₆₀) undergo eutectoid decomposition to produce a nano-chessboard structure, which is composed of two interleaved phases – the hard magnetic L1₀ phase (CoPt) and the soft magnetic L1₂ phase (CoPt₃). This material is expected to have potential for high density magnetic storage due to the exchange coupling between the hard and the soft magnetic phases. We use C_S-corrected Lorentz TEM in combination with micromagnetic simulations to characterize magnetic domain walls in Co_{40.2}Pt_{59.8} nano-chessboards. The output magnetization from the micromagnetic simulations is used to compute the magnetic phase shift of the electron wave using the Mansuripur algorithm [2,3]. Fresnel images and color integrated induction maps are simulated using the computed phase shift. A series of color maps is obtained for different beam tilts and magnetization states. Due to limitations in operating the LTEM in diffraction mode, the experimental Fresnel images are acquired at unknown sample orientations. Since a clockwise sample tilt inside the TEM is equivalent to an anti-clockwise tilt of the electron beam, the magnetic contrast in the experimentally obtained color map can be analyzed by comparing it with simulated color maps generated at different beam tilts.

Fig. 1(a) shows an experimental nano-chessboard TEM dark field image. The black tiles are the L1₀ phase and white tiles are the L1₂ phase. The easy axes of L1₀ tiles are labeled in the figure. Fig. 1(b) shows a Fresnel under-focus image. The zigzag or 'S' shaped magnetic domain walls are visible as white bright contrast in the under-focus image. The phase map and the color map of marked rectangular area are shown in Fig. 1(c) and Fig. 1(d), respectively. The color map represents the direction of

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integrated magnetic induction inside the sample. The overlaid magnetization vectors (white arrows) in Fig. 1(d) show the direction of magnetic induction in each of the L1₀ tiles. Fig. 2 shows the simulation results. The white magnetization vectors in Fig. 2(c) show a vortex-type pattern and the black magnetization vectors show an anti-vortex type pattern. The vortex type repetitive pattern can also be observed in the experimental color map. The type of magnetic domain arrangement is dependent on the relative orientation of the easy axis of magnetization with respect to the sample normal. Only the inplane component of magnetization gives rise to magnetic contrast. As the sample is tilted away from the zone axis [001], the in-plane magnetic component corresponding to each L1₀ tile changes and a non-symmetrical distribution of colors is seen in the color map. This asymmetry in the magnetic contrast can be seen in the simulated color map obtained at 15° beam tilt (Fig. 2(d)). The blue and yellow color contrast becomes more pronounced as the beam is tilted towards [010]. This is also evident in the experimental image. Future work involves extensive simulations combined with in-situ Lorentz imaging in order to understand exchange coupling in Co-Pt nano-chessboards.

References:

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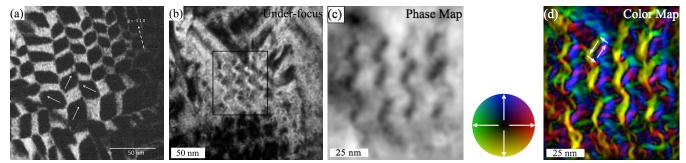


Figure 1. (a) Experimental nano-chessboard (-110) dark field image on [001] zone axis [4]; (b) Fresnel under-focus image ($\Delta f = 1 \mu m$) showing the S-shaped domain walls; (c) Phase reconstructed map; (d) Color induction map showing the direction of magnetic induction inside each of the black tiles.

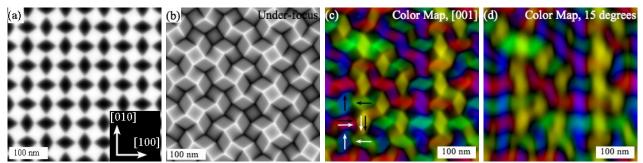


Figure 2. (a) Simulated 2-D nano-chessboard viewed along [001] direction; (b) Simulated Fresnel under-focus image showing zigzag domain wall contrast; (c) Color map at 0° beam tilt, [001] zone axis; (d) Color map at 15° beam tilt.