

Suspended and Bilayer Graphene Growth at Cu Grain Boundaries on Thin Film Cu

Lester F. Lampert and Jun Jiao¹

¹ Department of Mechanical & Materials Engineering and Department of Physics, Portland State University, Portland, OR, USA.

Although much progress has been made in regards to high-quality, monolayer graphene growth with copper (Cu) foils, Cu thin film-based graphene growth has seen little improvement largely due to the weak adhesion of Cu with the typical underlying substrate, SiO₂, and related evaporative losses of Cu. Therefore, the best route to high-quality copper thin film-based graphene growth is to optimize with an SiO₂ support and either drive the growth process to low temperature with a plasma process or optimize with a thermal growth system bearing in mind high temperature delamination and dewetting issues commonly experience with copper thin films. Here, we report the enhancement of graphene Raman signal intensity due to suspended graphene and bilayer formation at Cu grain boundaries.

Cu thin films were deposited onto SiO₂/Si wafers with DC sputtering until 1 μm thick and then placed directly into a cold-wall growth reactor optimized for graphene growth. Large Cu grains were achieved at heater temperatures of 780°C under H₂ flow for 5 mins. Electron backscatter diffraction (EBSD) mapping and scanning electron microscope (SEM) imaging were both performed with a Zeiss Sigma VP SEM equipped with a Nordlys EBSD camera. Optical inspection and Raman spectroscopy/mapping were performed with a Horiba HR800 Raman spectrometer. Raman maps were analysed with GRISP v1.2 [1].

After growth of graphene, Cu films formed into grains with a high density of twin boundaries as seen in Fig. 1a. Additionally, large separations between Cu grains of 0.5-1.0 μm formed (Fig. 1c). Within these separations bilayer graphene forms and stitches together the monolayer graphene highlighted in Fig. 1b. Post-transfer, graphene films crack only at the bilayer formed between Cu grain boundaries as indicated by optical inspection in Fig. 1d. Graphene nucleation density enhancement has been previously identified by Han et al. in accordance with Cu surface morphology [3]. In order to validate the growth of bilayer graphene between Cu grain boundaries, Raman mapping was performed across several Cu grain boundaries in Fig. 2a. Interestingly, suspended graphene was identified occasionally at Cu grain boundaries (Fig. 2b) with $I_{2D/G} > 9$ (Fig. 2e,h) due to reduced doping, resulting in lowered electron-electron scattering [2]. $I_{D/G}$ did not vary with significant spatial dependence as in Fig. 2c. G peak intensity, I_G , is localized to Cu grain boundaries (Fig. 2d) where $I_{2D/G} < 1$ (Fig. 2b) further indicating bilayer graphene formation at Cu grain boundaries. The 2D peak FWHM was larger at Cu grain boundaries indicated by a large relative shift to monolayer graphene as depicted in Fig. 2e. Suspended graphene demonstrated Raman spectra with near-neutral G peak positions having a slight red shift possibly indicating tensile stress present at the bilayer graphene grain boundaries (Fig 2g). [4]

References:

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- [3] GH Han, *et al* Nano Letters **11** (2011), p.4144.

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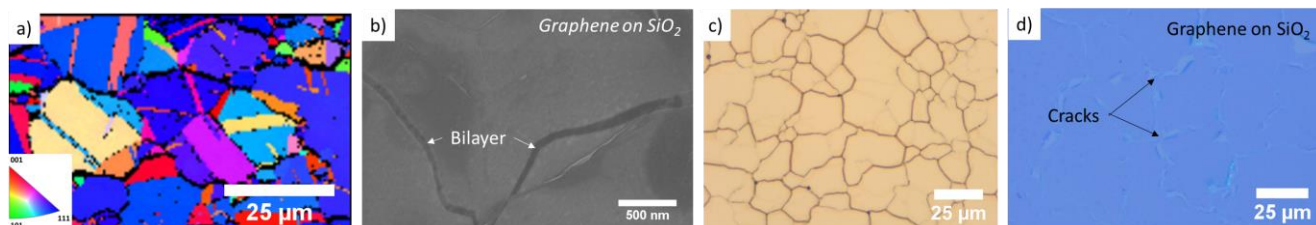


Figure 1. a) SEM EBSD inverse pole figure for Cu cubic orientation out of the sample plane (z), b) SEM image of the graphene transferred to SiO₂ demonstrating bilayer formation at Cu grain boundaries, c) optical brightfield image of as-grown graphene supported by Cu/SiO₂/Si, d) optical brightfield image of graphene transferred to SiO₂ with cracks forming at weak bilayer grain boundaries.

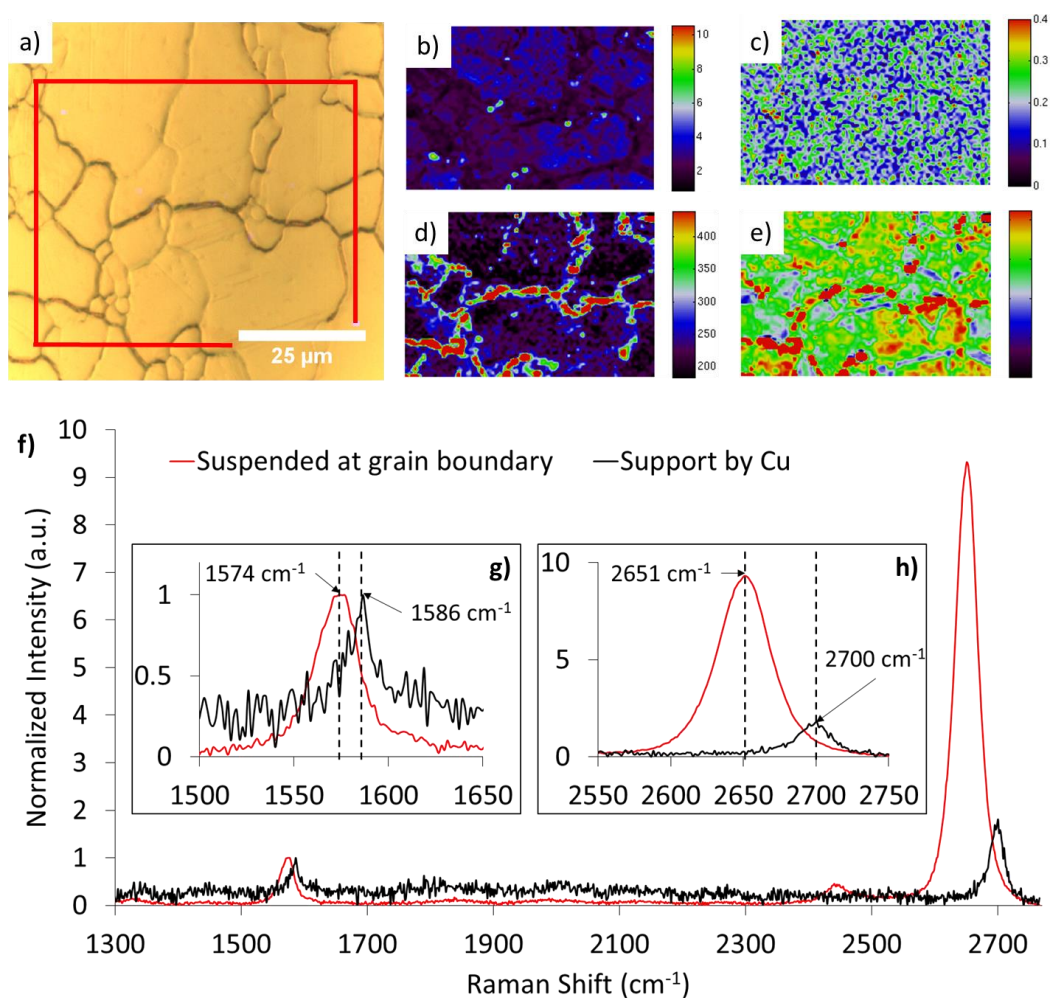


Figure 2. a) Optical brightfield image with region of interest marked for Raman maps, b-e) Raman maps for I_{2D/G}, I_{D/G}, I_G, and 2D relative FWHM, respectively, f) Raman spectra for suspended and Cu-supported graphene, g) G peak inset depicting relative shift, h) 2D peak inset depicting relative shift and enhancement of suspended graphene.